

Fabrication of Multi-material 3D Structures by the Integration of Direct Laser Writing and MEMS Stencil Patterning

Jeremy B. Reeves,^{*a} Rachael K. Jayne,^{*b} Lawrence Barrett,^c Alice E. White,^{bcd} and David J. Bishop^{abcde}

The construction of a complex, 3D optical metamaterial challenges conventional nanofabrication techniques. These metamaterials require patterning of both a deformable mechanical substrate and an optically-active structure with ~200nm resolution and precision. The soft nature of the deformable mechanical materials often precludes the use of resist-based techniques for patterning. Furthermore, FIB deposition approaches produce metallic structures with considerable disorder and impurities, impairing their optical response. In this paper we discuss a novel solution to this nanofabrication challenge -- the integration of direct laser writing and MEMS stencil patterning. We demonstrate a variety of methods that enable this integration and then show how one can produce optically-active, 3D metamaterials. We present optical characterization data on one of these metamaterials to demonstrate the viability of our nanofabrication approach.

1. Introduction

The construction of 3D optical metamaterials is a significant nanofabrication challenge. One typically needs to pattern both the mechanical substrate as well as the optically-active elements, each with deep sub-micron resolution, and then align them relative to each other, also with deep sub-micron precision. A powerful approach for fabricating the mechanical layer is direct laser writing (DLW) via two-photon polymerization (TPP). DLW provides a method for the fabrication of complex 3D shapes with arbitrary geometries and minimum feature sizes of ~200 nm. This technique enables rapid prototyping of virtually any nanoscale structure, and has been used to produce nanolattices, sensors and actuators, photonic crystals, and metamaterials.¹⁻⁵ Currently, the materials compatible with DLW are largely limited to dielectric polymers.

In many cases, DLW-fabricated structures often serve as a starting point for multi-material integration. Coating DLW structures with a desired final material including ceramics⁶ and metals^{2,3,7,8} allows for the creation of structures with geometries that are not practical using planar lithographic methods. Fabricating mechanical,⁹ magnetic,¹⁰ and optical metamaterials^{1,11} with properties derived from the microscale 3D geometries enabled by DLW remains a fruitful line of research. So far, much of this work has been limited to conformal coating of DLW structures. Methods for selective patterning of DLW structures or integration

of non-polymer materials have remained somewhat limited. While procedures based on two-photon reduction and metal-ion doped polymers have produced useful devices and structures,¹²⁻¹⁴ combining DLW with established patterning and deposition techniques will allow for complex multi-material and material-agnostic fabrication processes. Applications will include 3D optical metamaterials as described below and the construction of 3D scaffolds with embedded biosensors for tissue engineering.

Previously, we reported the selective patterning of metallic split-ring resonators (SRRs) on a mechanically flexible 3D structure¹⁵. Despite the resolution achieved and the underlying complex 3D structure of the device, the surface on which metallic patterning was performed was 2D. Here, we demonstrate that extending the stencil lithographic technique to 3D multilayer microstructures allows for the creation of optical metamaterials with a response enhanced by the custom-designed 3D architecture.

We present a method for selective patterning of metallic structures onto 3D polymer microstructures. Complex 3D structures like those shown in Figure 1 (a,b) are printed using DLW and then patterned with metallic features using microelectromechanical systems (MEMS) stencils using the procedure schematically outlined in Figure 1 (c-e). By implementing multiple stencil lithography steps, in principle it is possible to build each layer out of independent patterns and materials. The stencil process frees the design of 3D-structures from constraints associated with geometries and chemistries of traditional planar lithography or two-photon reduction for the fabrication of metallic features. Here, we use electron-beam evaporation to pattern elemental metal, but other materials and physical vapor deposition (PVD) methods can also be used with the technique including processes

^a Department of Electrical and Computer Engineering, Boston University, Boston, Massachusetts, 02215

^b Department of Mechanical Engineering, Boston University, Boston, Massachusetts, 02215

^c Division of Materials Science and Engineering, Boston University, Boston, Massachusetts, 02215

^d Department of Physics, Boston University, Boston, Massachusetts, 02215

^e Department of Biomedical Engineering, Boston University, Boston Massachusetts, 02215

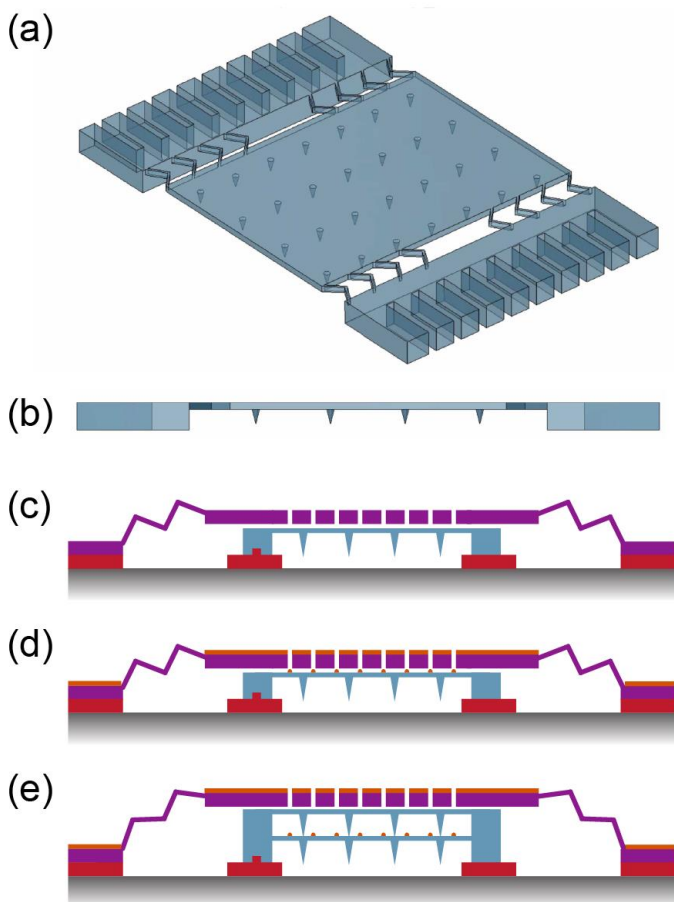


Figure 1: Schematic of the multistep process for fabricating 3D metal-polymer structures. (a) Oblique and (b) side view of the DLW-printed structure. (c) Once the stencil pattern has been fabricated in the mask (purple) and the first polymer structure (blue) has been printed onto the MEMS substrate, the sacrificial oxide layer is removed and the mask is positioned on top of the structure. (d) Next, metal is evaporated through the mask using electron beam PVD. The polymer structure is to its initial position. (e) Another polymer layer is fabricated. The positioning and deposition processes are repeated.

for depositing organic molecules such as organic vapor jet printing.¹⁶

2. Results and Discussion

MEMS mechanical elements are used both to position the mask and to manipulate the entire 3D structure. The devices used in this study are shown in Figure 2. All MEMS devices used here are fabricated using the PolyMUMPS process at the MEMSCAP foundry.¹⁷ PolyMUMPS allows for two mechanically movable layers, referred to as poly-1 (P1) and poly-2 (P2). Generally, these layers make up the mask (P2) and carrier

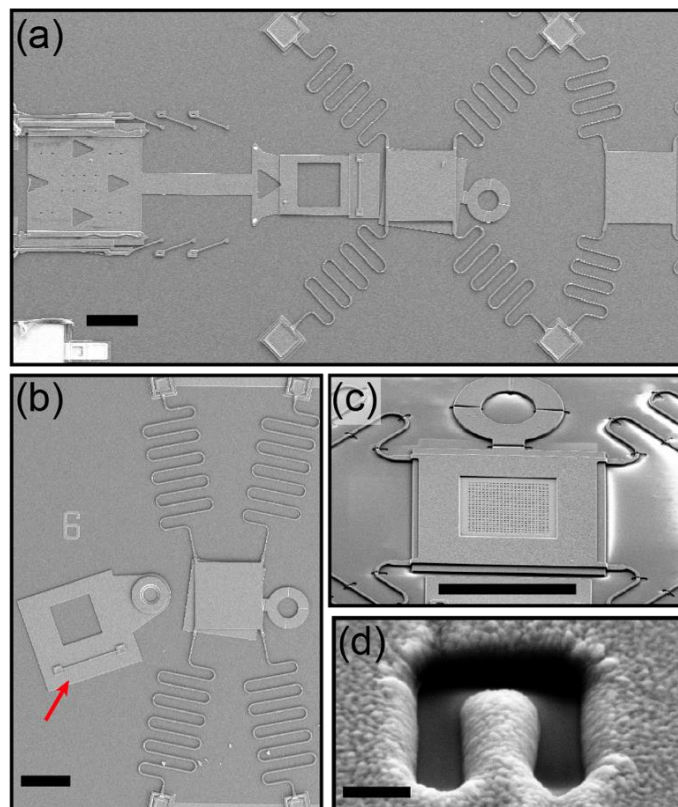


Figure 2: MEMS devices and FIB-processing of MEMS stencil masks. (a) A sliding shuttle design with two spring tethered masks for multiple mask patterns (b) Rotating plate design with raised "dumbbell" structure (arrow). (c) FIB patterned stencil area on a mask plate. (d) Single repeated element of stencil mask pattern. To maximize pattern resolution, the polysilicon is thinned in the FIB to about 300 nm. (Scale bars represent 100, 100, 100, and 0.5 μm respectively.)

(P1) layer for the fabrication of micro-patterned multilayer DLW structures. The MEMS stencil mask layer is similar to that of Atomic Calligraphy¹⁸, where masks are suspended by soft silicon springs to enable better alignment with 3D structures. The carrier layer is capable of acting as a mobile substrate for the 3D-structures, and can be moved independently of the mask layer. The inclusion of this mobile substrate is important for repositioning the 3D structures under various stencil masks while also providing adequate contact to the base die, which is crucial for DLW printing alignment.

We explore the utility of two possible carrier layer designs in multilayer printing. The first, a sliding shuttle seen in Figure 2 (a), can be aligned with two separate mask plates to provide the movable substrate for the 3D structure. This allows positioning of the shuttle under the first mask to pattern the first layer of the DLW structure. After a metallic deposition step, the shuttle is returned to its initial position and a second DLW step prints another polymer

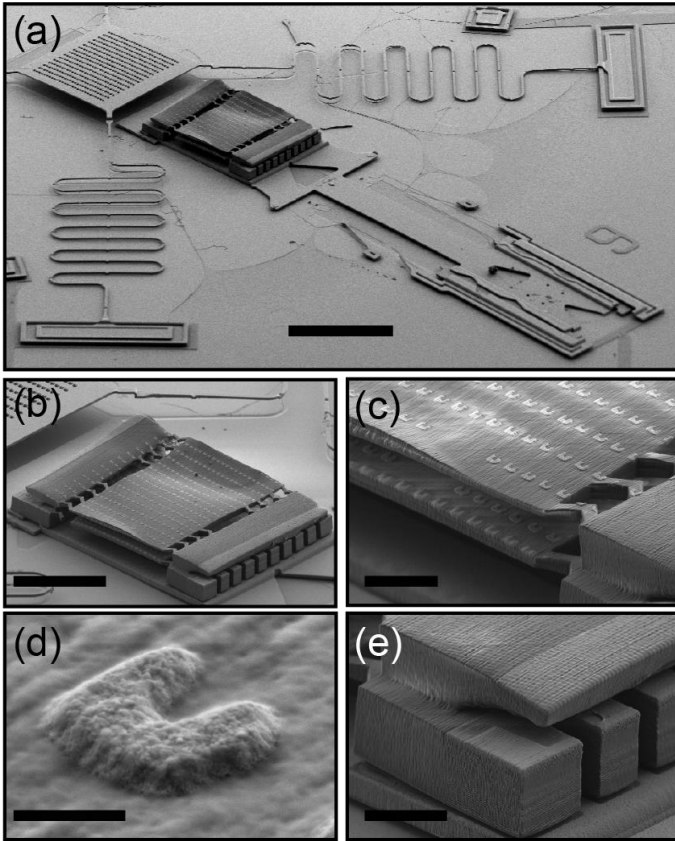


Figure 3: Fabricated multilayer structures. (a) The patterned, multilayer structure positioned back in the initial shuttle position after a second aluminum deposition step. (b) The double layer structure. (c) Close-up of the separated, patterned layers. (d) Close-up of an individual split-ring structure. (e) Close-up of the joint between the first and second DLL layers showing good polymer-polymer adhesion. (Scale bars represent 100, 50, 10, 1, and 10 μm respectively.)

layer on top of the first. The shuttle and mask device layers are manipulated using a probe station. This sliding shuttle design allows the use of two separate masks, so each layer can be patterned independently.

A second design, shown in Figure 2 (b), relies on a rotating hinge to reposition a DLW-fabricated structure beneath a mask. By simply rotating the carrier layer through ~ 180 degrees, the 3D structure is positioned beneath the mask. Manipulation of the mask-hinge system is easier compared to the sliding shuttle design, because the mobile substrate is fixed to the MEMS substrate. The specific design addressed here is limited to a single mask pattern and could be used to create repeating stacks of independent layers. In principal, this design could accommodate a variety of masks if the MEMS tether and mask are designed appropriately.

In order to produce stencil masks with features smaller than the 2 μm critical dimension of the foundry process,

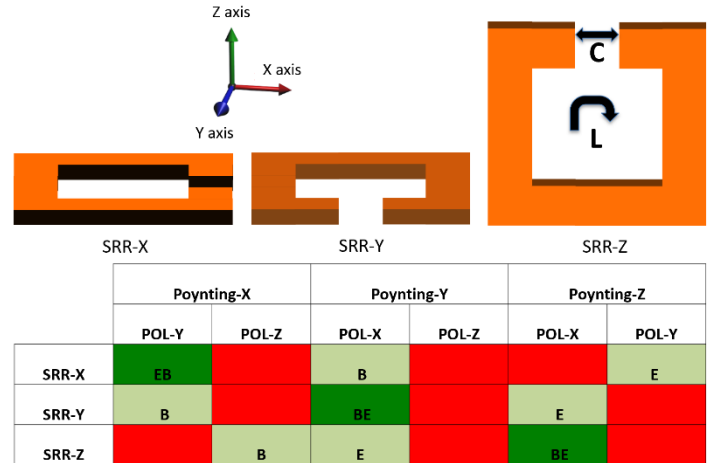


Figure 4: Shown are the 18 combinations of SRR, Poynting vector and polarization orientations and the coupling of light to the magnetic dipole resonance. In the nomenclature used here SRR-X,Y,Z refers to the orientation of the SRR as shown above, Poynting-X,Y,Z refers to the direction of the Poynting vector of the light incident on the SRR and POL-X,Y,Z refers to the linear polarization of that light. The table indicates the coupling between the incident light and the SRR. Green indicates strong coupling (to both E and B fields), light green indicates weak coupling (only E or B field alone), and red indicates no coupling.

patterns were milled into the foundry-fabricated mask plates using focused-ion beam (FIB). This allows for stencil patterns with feature sizes below 0.5 μm with resolution of ~ 200 nm. As seen in Figure 2 (c-d), the 1.25- μm -thick P2 plate is thinned to about 300 nm before a stencil pattern is milled through the plate. The resulting patterns have attained features well below 1 μm .

The devices arrive from the foundry encapsulated in a sacrificial oxide. FIB patterning and the first DLW printing is performed directly on the unreleased MEMS die. DLW printing is carried out using the Nanoscribe Photonics Professional GT with a 63x, NA=1.4 immersion objective and IP-Dip photoresist. After processing, the

device is then released by submersion in hydrofluoric acid and undergoes critical point drying. The raised dumbbell-shaped structures on the shuttle and hinge-plate in Figure 2 (a,b) provide a strong attachment point for the 3D structures, improving device yield of the release step to nearly 100%. A probe station is used to position the DLW structures below the patterned stencil masks and PVD of the selected material is performed. After deposition, the carrier layer is returned to its initial position. For subsequent layers, the process repeats. Although the device is now released, the shuttles are repositioned in the guiding rails after deposition to prevent them from floating away when the MEMS device is immersed again in photoresist for the subsequent DLW printing step.

DLW structure designs are shown in Figure 1 (a,b). These structures have several key features. In order to minimize surface roughness which is important for metamaterial fabrication, the flat central area is printed with a small hatch value, which means that for each subsequent line, the laser focus overlaps significantly with the previous line. To prevent the printed layer from curling up and detaching from the surface due to post-polymerization shrinkage, deformable polymer springs attach the structure to the anchors on either side. Furthermore, the side anchors have “teeth” to maximize surface area, thereby enhancing contact/cross polymerization between the two printed layers. With the central layer suspended, small posts on the underside of the flat sheet reduce stiction. The spacing between the two layers is limited by resist bubbling caused by local heating that arises when patterned metallic elements are irradiated by the laser focus. Metallic elements can be completely encapsulated by the DLW polymer material, but there is a trade-off between the polymerization power that can be used before local bubbling near the surface of the metal, and structure shrinkage that ultimately warps the final printed layer. In this study, we found that a spacing of $\sim 2 \mu\text{m}$ is sufficient to prevent bubble formation.

The results of all process steps are shown in Figure 3, which shows a device with two layers of aluminum-patterned IP-Dip. The movable shuttle has been replaced to its initial position. As seen in Figure 3 (b-c), both layers have identical SRR patterns deposited on their surface and the separation between the layers is $\sim 2 \mu\text{m}$. Because the surface of the 3D structure to be patterned is flat, the mask is in close contact with the surface, allowing for accurate reproduction of the stencil pattern in aluminum on the surface. The SRR in Figure 3 (d) shows straight sidewalls and negligible surface diffusion, which indicate that geometric spread and diffusion typically associated with stencil lithography can be controlled. Figure 3 (e) indicates good adhesion between the first and second DLW-printed layers.

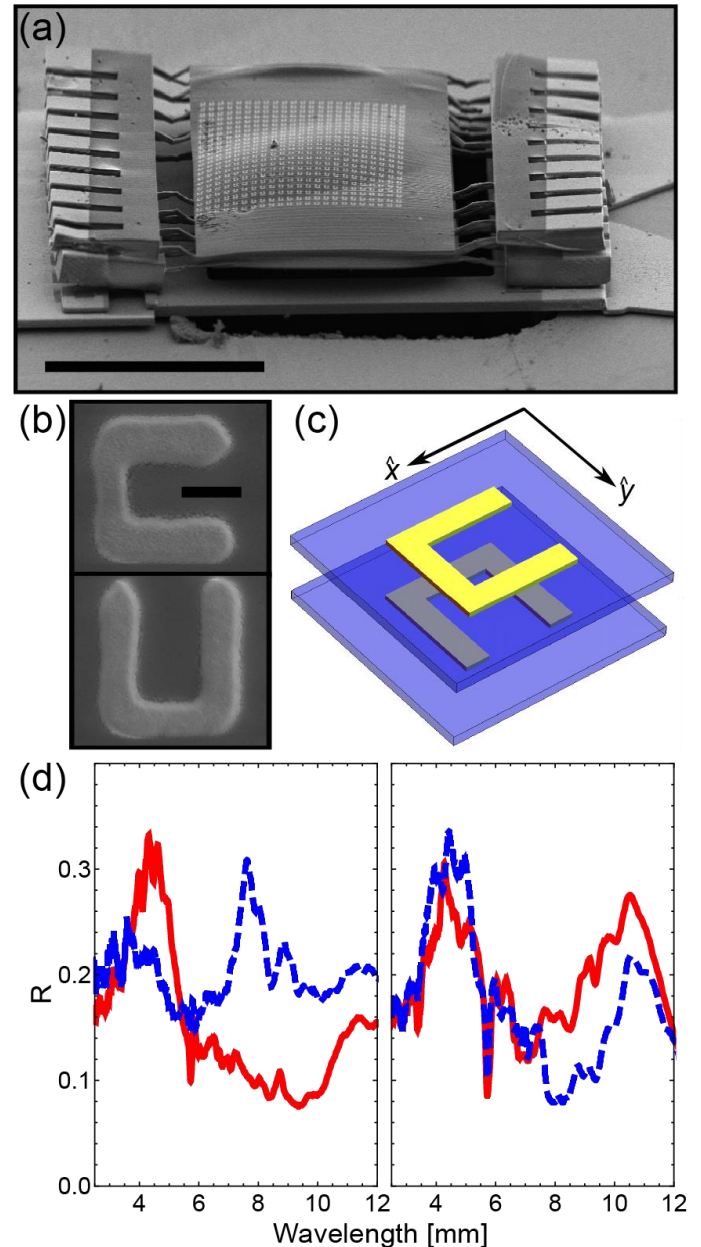


Figure 5: Fabricated multi-material structures with optical response in the mid-infrared. (a) The fabricated multilayer structure. (b) SEM image of a gold split ring resonator from the first layer (top image) and the second layer (bottom image) of the two-layer stack. A single layer device consists only of the split rings on the first layer. (c) Schematic of the multi-layer structure with relative split ring orientations. (d) Reflection spectrum of a single (left) and double (right) layer structures showing the fundamental magnetic and electric resonances of the split ring structures. Spectra for light polarized along \hat{y} (\hat{x}) are plotted in solid red (dashed blue). The scale bars in (a) and (b) correspond to 100 and $0.5 \mu\text{m}$ respectively.

As shown in Figure 4, full design flexibility in optical metamaterials requires the ability to orient the SRRs in any of three directions in space. In general, there are two main fundamental resonances, magnetic dipole with current circulating around the SRR and electric dipole with current running across the arms of the SRR. The magnetic dipole is excited when the electric field is directed across the gap (in the y -direction for SRR- x or in the x -direction for SRR- y in Figure 4), or when the magnetic flux penetrates the ring, which occurs when the electric field and Poynting vector are in the plane of the SRR. The electric dipole will be excited anytime the electric field is parallel to the SRR arms (along x , y , or z for SRR- x , SRR- y , or SRR- z , respectively). Optical response from the SRR depends on the specific relative orientations between the incident optical field and the SRR. The table in Figure 4 shows the 18 combinations of SRR, Poynting vector and polarization orientations and the coupling of light to the magnetic dipole resonance. A full discussion of the geometry of SRR orientations relative to the Poynting vector of incident light is not appropriate here, but overall, the ability to engineer a strong response to any specific configuration of incident light requires that one be able to rotate the SRRs around all three axes.

In order to characterize the quality of the metallic patterns fabricated on each layer in the multi-deposition experiments, we fabricate a two-layer structure with 23×23 arrays of gold $1.2 \mu\text{m}$ square split-ring resonators on each layer, as seen in Figure 5 (a). The SRR orientation is rotated by 90 degrees between the two layers in order to distinguish their optical responses. The split rings in Figure 5 (b) are fabricated using the device in Figure 2 (a), each mask plate having an array of differently oriented resonators. We note that the structure demonstrated in Figure 5 (a) has a slight curvature to the upper surface. This can be attributed to bubbles, which were generated in the resist during the second DLW step, moving the polymerized layers as the structure prints. These bubbles can be prevented by suspending the adjacent layer by at least $2 \mu\text{m}$ above any patterned regions. We further note that because both masks are exposed to the atomic flux during subsequent evaporations, the upper layer has slightly narrower features due to partial filling of the mask pattern by the deposited metal.

We measure the optical response of this structure using a Fourier-transform infrared spectrometer with attached microscope in reflection mode (Bruker Hyperion 1000). Each structure is illuminated by light polarized along the \hat{x} or \hat{y} axes. The single-layer device (represented by only the lower layer of the two-layer devices) exhibits two resonances, as seen in Figure 5 (d), corresponding to the fundamental electric and magnetic dipole resonances of the SRRs around $4 \mu\text{m}$ and $8 \mu\text{m}$ respectively.¹⁹ As the Poynting vector is perpendicular to the planes containing

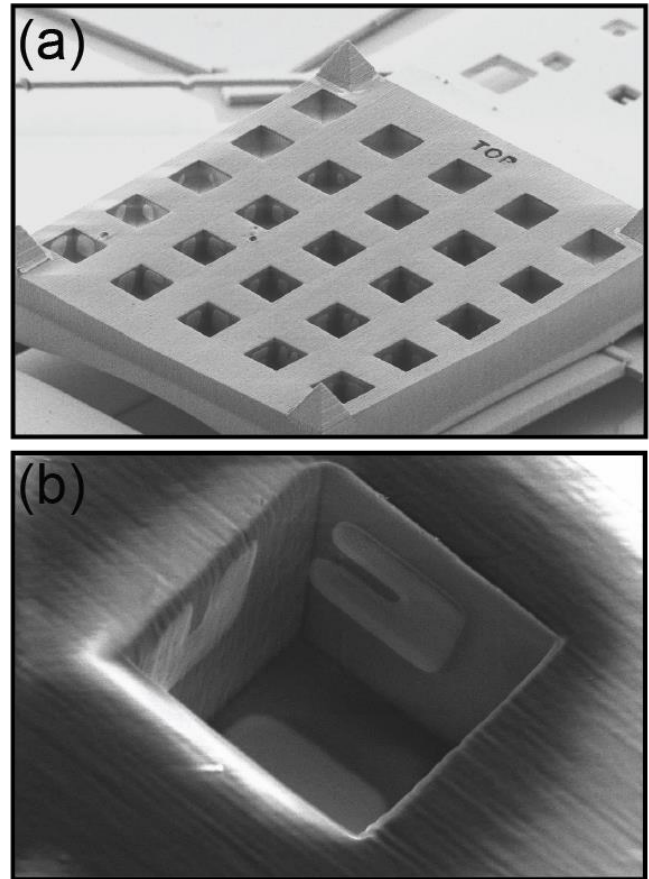


Figure 6: Shown are results using the approach described in Figures 1,2,3 to produce a structure where the SRR's are oriented in all three directions, like the sketch in Figure 4. The substrate structure was built using the DLW tool and alignment posts (visible at the corners of the structure) were used to align the MEMS mask as shown in detail in Figure 2 (a).

the SRRs, each resonance driven by the electric field is individually excited by orthogonal polarizations of the infrared radiation. The multilayer structure demonstrates similar resonances; however, the spectra do not display the same polarization sensitivity as is seen in the single layer device. Both layers of the 3D-structure contribute to the optical response. While IP-Dip is an excellent resist for high resolution DLW fabrication, its acrylate formulation contributes to strong absorption features around $5 \mu\text{m}$ and $8 \mu\text{m}$.¹⁵ These features are particularly noticeable in the multilayer structure since the light must pass through the IP-Dip structure after being reflected.

The data in Figure 5 clearly demonstrates that one can use this integrated approach to build 3D optical metamaterials. The use of an epoxy-based resist such as SU-8 may avoid some of these sharp infrared absorptions while still allowing high resolution DLW.^{20,21} Avoiding such lossy features or designing the metallic pattern for resonances in other spectral regions will lead to the fabrication of high performance metamaterials and surfaces.

Having freedom to fabricate the SRR in any of the configurations in Figure 4 could lead to metamaterials with isotropic, polarization-insensitive responses. Using the methods described here, a DLW structure was patterned using angled deposition through a MEMS stencil (Figure 6), demonstrating full flexibility in how the SRRs are oriented. Proper registration between these two components is achieved through the inclusion of self-alignment posts on the DLW-fabricated 3D structures and corresponding alignment apertures on the MEMS stencil. The alignment posts on the mechanical layer determine the registry of the MEMS mask layer to a precision of a few hundred nanometers, thus eliminating the need for any kind of optical alignment step. One can write nanoscale patterns on all five surfaces of the wells. The foreshortening effects on the vertical sidewalls are predictable and can be corrected for in the design of the apertures in the stencil.

3. Conclusions

We have demonstrated a method integrating DLW and MEMS stencil lithography to build multi-material 3D structures. DLW 3D structures can be patterned with metallic nanostructures to produce optical metamaterials with useful geometries that are nearly polarization insensitive. Additional studies will investigate the possibility of using non-planar DLW substrates in tandem with techniques like angled deposition to enhance the complexity of structures that can be fabricated using this approach.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was funded by Boston University Photonics Center, Boston University College of Engineering, the DARPA Atoms to Product (A2P) Program/Air Force Research Laboratory (AFRL) contract no. FA8650-15-C-7545, NSF-CMMI-1361948 and the NSF ERC Program under Cooperative Agreement No. EEC-1647837.

Notes and references

- 1 M. S. Rill, C. Plet, M. Thiel, I. Staude, G. Von Freymann, S. Linden and M. Wegener, *Nat. Mater.*, 2008, **7**, 543–546.
- 2 F. Formanek, N. Takeyasu, T. Tanaka, K. Chiyoda, A. Ishikawa and S. Kawata, *Opt. Express*, 2006, **14**, 800.
- 3 V. Mizeikis, S. Juodkazis, R. Tarozaitė, J. Juodkazyte, K. Juodkazis and H. Misawa, *Opt.*

- 4 *Express*, 2007, **15**, 8454–8464.
- 5 F. Klein, T. Striebel, J. Fischer, Z. Jiang, C. M. Franz, G. Von Freymann, M. Wegener and M. Bastmeyer, *Adv. Mater.*, 2010, **22**, 868–871.
- 6 M. Deubel, G. Von Freymann, M. Wegener, S. Pereira, K. Busch and C. M. Soukoulis, *Nat. Mater.*, 2004, **3**, 444–447.
- 7 D. Jang, L. R. Meza, F. Greer and J. R. Greer, *Nat. Mater.*, 2013, **12**, 893–8.
- 8 K. M. Huang, S. C. Tsai, Y. K. Lee, C. K. Yuan, Y. C. Chang, H. L. Chiu, T. T. Chung and Y. C. Liao, *RSC Adv.*, 2017, **7**, 51663–51669.
- 9 N. Takeyasu, T. Tanaka and S. Kawata, *Appl. Phys. A Mater. Sci. Process.*, 2008, **90**, 205–209.
- 10 L. R. Meza, A. J. Zelhofer, N. Clarke, A. J. Mateos, D. M. Kochmann and J. R. Greer, *Proc. Natl. Acad. Sci. U. S. A.*, 2015, **112**, 11502–7.
- 11 G. Williams, M. Hunt, B. Boehm, A. May, M. Taverne, D. Ho, S. Giblin, D. Read, J. Rarity, R. Allenspach and S. Ladak, *Nano Res.*, 2018, **11**, 845–854.
- 12 J. K. Gansel, M. Thiel, M. S. Rill, M. Decker, K. Bade, V. Saile, G. Von Freymann, S. Linden and M. Wegener, *Science (80-.)*, 2009, **325**, 1513–1515.
- 13 X. M. Duan, H. B. Sun, K. Kaneko and S. Kawata, *Thin Solid Films*, 2004, **453–454**, 518–521.
- 14 S. Tabrizi, Y. Cao, B. P. Cumming, B. Jia and M. Gu, *Adv. Opt. Mater.*, 2016, **4**, 529–533.
- 15 S. Tabrizi, Y. Y. Cao, H. Lin and B. H. Jia, *Sci. China Physics, Mech. Astron.*, 2017, **60**, 1–17.
- 16 J. B. Reeves, R. K. Jayne, T. J. Stark, L. K. Barrett, A. E. White and D. J. Bishop, *Nano Lett.*, , DOI:10.1021/acs.nanolett.7b05042.
- 17 United States Patent and Trademark Office, US7431968B1, 2008, Oct, 26.
- 18 D. Koester, A. Cowen and R. Mahadevan, *MEMSCAP Inc.*
- 19 M. Imboden, H. Han, J. Chang, F. Pardo, C. A. Bolle, E. Lowell and D. J. Bishop, *Nano Lett.*, 2013, **13**, 3379–3384.
- 20 C. Rockstuhl, F. Lederer, C. Etrich, T. Zentgraf, J. Kuhl and H. Giessen, *Opt. Express*, 2006, **14**, 8827.
- 21 D. B. Fullager, G. D. Boreman and T. Hofmann, *Opt. Mater. Express*, 2017, **7**, 888.
- 22 S. Juodkazis, V. Mizeikis, K. K. Seet, M. Miwa and H. Misawa, *Nanotechnology*, 2005, **16**, 846–849.